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--- SYNTHESIS AND CHARACTERIZATION OF ZINC OXIDE THIN FILMS BY SOL-GEL METHOD ON GLASS SUBSTRATE

¹E.ELANCHEZIAN and ² P.VIJAYALAKSHMI

 1.2 PG & Research Department of Physics ^{1,2}Sri Vidya Mandir Arts and Science College, Katteri, Uthangarai.

ABSTRACT

Nanocrystalline ZnO thin films have been fabricated by a multi-step sol-gel method using spin coating technique. Zinc acetate dihydrate, 2-methoxyethanol and monoethanolamine were used as a starting material, solvent and stabilizer, respectively. X-Ray Diffraction (XRD) and Scanning Electron Microscopy (SEM) were employed to characterize structure and morphologies of the asdeposited samples. The results of XRD and SEM showed that the heat treatment conditions, final rotational (spinning) speed, fume exhaust and precise control of concentration of reactants (precursor and solvent used) strongly affect the crystallographic orientation and morphology of the resultant ZnO films. The XRD pattern showed that the ZnO films formed the preferred orientation along c-axis and the grain size is 16nm for the samples. Only one peak corresponding to the (002) plane at $2\theta = 34.34^{\circ}$ appears on the diffractograms. The as-deposited films had a transparency of greater than 80% in the visible-near IR region from 400 nm – 800 nm. The optical band gap energy and thickness were calculated to be 3.296 eV and 266 nm respectively.

1. INTRODUCTION

ZnO is an inexpensive n-type of II-VI semiconductor compound, which has technical applications such as photo-catalysts thin film gas sensors [2], varistors [3], light emitting diodes [4], spintronic devices [5], nanolasers [6]. ZnO thin films have also been widely used as surface acoustic wave (SAW) device and film bulk acoustic resonator (FBAR) because of its excellent piezoelectric properties [7,8]. Its stability in hydrogen plasma [9] culminated in its exhaustive use as window layer for polycrystalline solar cells [10-12] and silicon thin film solar cells [13-15]. Moreover, ZnO has large band gap 3.37 eV, large excitonic binding energy 60 meV and high carrier mobility at

hexagonal wurtzite crystal structure with unit cell a = 3.253 Å and c = 5.215 Å due to their unique optical, electrical and semi-conducting properties, ZnO thin films form the mainstay of the electronics industry and the cornerstone of modern technology. Nanostructures of ZnO are fabricated using various thin film techniques as sputtering [16], spray pyrolysis [17] metalorganic chemical vapour deposition (MOCVD) [18], molecular beam epitexy (MBE) [19], pulsed laser deposition (PLD) [20] and the solgel process [21-27]. Sol-gel method is widely adopted for the fabrication of transparent and conducting oxide due to its simplicity, safety, no need of costly vacuum system and hence cheap method for large area coating. The sol-gel

room temperature. ZnO is composed of

process also offers other advantages such as high surface morphology at low crystallizing temperature, the easy control of chemical components and fabrication of thin film at low cast for elucidating the structural and optical properties of ZnO thin film. The fabrication of doped and un-doped ZnO thin films by the solgel process has already been reported by many researchers [24-27]. However, up to date, multiple deposition steps have been generally necessary to produce thin films of high-quality. In order to obtained high-quality films, a typical multi-step deposition process has been demonstrated.

In the present work, nanoycrystalline ZnO films have been produced by the sol-gel method using zinc acetate precursor and their surface morphologies, preferential orientation and optical properties were studied in detail.

2. MATERIALS AND METHODS

2.1. ZnO Sol-Gel Preparation

The sol-gel synthesis and thin film process arrangement is shown schematically in Figure 1. Zinc acetate dehydrate (Zn(CH2COO)2·2H2O) (ZAD), 2-methoxyethanol (CH3O(CH2)2OH) (2-ME) and monoethanolamine ((HOCH2CH2) NH2) (MEA) were used as starting material, solvent and stabilizer, respectively. The mixture was confected to 0.5mol·L–1 and stirred magnetically. After stirring for 30minutes at 60˚C, MEA was added drop by drop underconstant stirring. The resultant solution was stirred for 90 minutes. to yield a colorless, homogeneous and transparent solution. The solution was aged for 72 hours at room temperature in order to make it more glutinous. The mo- lar ratio of MEA to ZAD was maintained at 1:1. Prior to the coating of the film, the glass substrates weresonicated or washed with detergent, and then cleaned in methanol and acetone for 5 minutes each.

Afterwards, substrates were rinsed with distilled water and then dried in hot air. The aged solution was dropped on glass substrates which were rotated at 2000 rpm for 30 seconds. The as-depo- sited films were then pre-heated at 200˚C for 10 minutes into a furnace to evaporate the solvent and remove organic residuals. This spinning to preheating procedure was repeated eight times. After the deposition of final layer, films were calcined in air at 400˚C for 1hour to ensure that all organic species were expelled from the films.

2.2. Material Characterization

X-ray diffraction (XRD) was used for the physical structure (crystalline structure and microstructure) of the ZnO thin films. XRD patterns were obtained with a MRD-Single Scan diffractometer with Cu Ka $(\lambda = 1.54056\text{\AA})$ radiation and scanning range of 2θ set between 20˚ and 80˚. During the measurement, the current and the voltage of XRD were maintained at 20 mA and 36 KV respectively, and scan speed was 4˚/min. The surface morphology of films was evaluated using Scanning electron microscopy (FEG-SEM, Nova-400). The thickness of the film was measured by a surface profilometer (AMBIOS). The transmission spectra of the films were measured by a double-beam ultraviolet/visible (UV-4100) spectrophotometer with a wavelength range 200 nm - 800 nm and the optical band gap was measured from the transmission spectra.

2.3. Growth Mechanisms

Sol-gel technique basically involves solution preparation which usually contains the metal alkoxide compounds M (OR)X where M is metal and R is alkyl radical or metal get a stable homogenous solution. The explanation of the formation of ZnO crystals in [28] enlighten that there are two possible ways of describing the growth of ZnO crystal; Aggregation and

Ostwald Growth (ripening). As soon as the smallest stable molecular clusters (they may be unit cells) are formed, they rapidly combine to give the next most stable aggregate. The primary aggregates would further rapidly combine to give the next most stable secondary aggregate and so on. The authors observed that the primary clusters were stable aggregates and would be the result of rapid aggregation rather than a result of Ostwald growth and concluded that the Ostwald mechanism should be considered as only one possible approach to the formation of bulk materials. Tokumoto et al. [29] reported that the formation of ZnO colloidal particles in an alcoholic solvent consists of two stages. During the early stage of phase transformation, small olygomers are continuously formed. At advanced stages, the aggregation of the olygomers leads to crystalline wurtizite, the primary colloidal particles. The primary particles then aggregate and form a third family, the secondary colloidal particles.

The growth of the colloidal particles is a stepped, discontinuous process indicating that the predominant mechanism of aggregation is heterogeneous coagulation. This mechanism of formation and growth leads to a hierarchical structure.

3. RESULTS AND DISCUSSION

3.1. Structural Analysis of Nanoycrystalline Films

Figure 2 depict the X-ray diffraction (XRD) pattern of the crystal structure and orientation of the nanocrystalline ZnO thin film deposited on glass substrate using spincoating at 2000rpm, pre-heated at 200˚C and annealed in air at 400˚C. From the XRD pattern, one can clearly observe a diffraction peak at $2\theta = 34.34^{\circ}$. Strong preferential growth is observed along c-axis i.e. (002) plane, suggesting that the prepared ZnO nanocrystals have the wurtzite structure. Sumetha Suwanboon [30] proposed a qualitative idea for the formation mechanism of the preferential oriented thin films could be the

minimization of the surface free energy of each crystal plane and usually films grows so as to minimize the surface energy. Due to the minimization of surface energy, heterogeneous nucleation readily happens at the interface of film and substrate.

3.2. Morphological Analysis of Nano-Crystalline Films.

The surface topography of thin film is very important tool to investigate microstructure of the films. Scanning Electron Microscopy (SEM) micrograph of ZnO thin film is shown in Figure 3(a) with presence of tightly packed grains. The nanocrystals are regularly distributed on the glass substrate and crystallite size is approximately 16 nm. It canbe seen from Figure 3(a) that the small grains made a smooth and transparent surface similar to those observed by other author [32]. The average film thickness is about 266nm. Figure 3(b) delineated energy dispersive scattering (EDS) of the above representative film. It may beseen that besides the characteristics peaks of Zinc and Oxygen, peaks arising out of the substrate Manganese, Gallium, nSilicon and Gold are also available.

3.3. Optical Properties of Nanocrystalline ZnO Films

Figure 4 shows the optical transmittance spectrum of nanocrystalline ZnO thin films annealed at 400˚C in air for 1 hour using UVvisible region from 200 nm - 800 nm. The transmittance is over 80% in the visible region from 400 nm to 800 nm for all the samples [32]. Sharp absorption edge is located at 380 nm which is due to the fact that the ZnO is a direct band gap semiconductor. The corresponding optical band gap of ZnO thin film is estimated by extrapolation of the linear relationship between (αhυ)2 and hυ according to the equation [33]

$$
a\mathbf{h}\mathbf{v} = \mathbf{A} \left(\mathbf{h}\mathbf{v} \mathbf{-Eg} \right)^{1/2} \tag{6}
$$

where α is the absorption coefficient, hu is the photon energy, Eg is the optical band gap and A is a constant. Figure 5 depicts the plot of $(\alpha h\nu)^2$ versus photon energyhυ. The values of the direct optical band gap Eg are calculated from the intercept of $(ahu)^2$ vs hv curve had alsobeen plotted. The presence of a single slop in the plot suggests that the film has direct and allowed transition. The band gap value of ZnO thin film is found to be 3.296 eV which is slightly smaller than that of bulk ZnO (3.37 eV). This difference is due to the fact that the values of band gap Eg depend on many factors e.g. the granular structure,the nature and concentration of precursors, the structural defects and the crystal structure of the films. Moreover, departures from stoichometry form lattice defects and impurity states. Dinghua Bao et al. [34] reported that the band gap difference between the thin film and crystal is due to the grain boundaries and imperfections of the polycrystalline thin films. D. L. Zhang et al. [35] reported that this band gap difference between the film and bulk ZnO is due to the grain boundary, the stress and the interaction potentials between defects and host materials in the films.

4. CONCLUSION

In this study, we have grown nanocrystalline ZnO thin films on glass substrate by a multi-step sol-gel technique using spin coating. The structural, morphological and optical properties were investigated. The sol-gel method is a relatively simple technique; there are many factors which affect the quality of the films. We have optimized different parameters to obtain a good crystalline structure of ZnO film with intense and sharp peak. According to XRD results, the as-deposited films exhibited a hexagonal wurtzite structure with (002) preferential orientation after annealing at 400˚C in air ambiance for 1 hour. The XRD pattern consists of a single (002) peak which occurred due to ZnO crystals and grows along the c-axis. The grain size and thickness of the films are estimated to be 16 nm and 266 nm. SEM micrograph of ZnO thin film shows that the small grains made a smooth and transparent surface. The optical transmittance is over 80% in the wavelength range from 400 nm - 800 nm

and the energy band gap is found to be 3.296 eV. Zinc oxide crystallites with preferential (002) orientation are desirables for applications where crystallographic anisotropy is a prerequisite e.g. UV diode lasers, piezoelectric surface acoustic wave or acousto-optic devices

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